

# Ab initio calculations at interfaces

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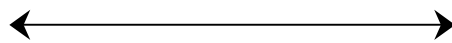
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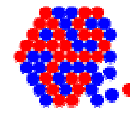
*Northern Ireland UK*

# Theme

Atomistic Calculations



Gibbs' thermodynamics



<http://titus.phy.qub.ac.uk>

See website for publications

# Thanks!

ASG staff:

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EC: TMR, HCM

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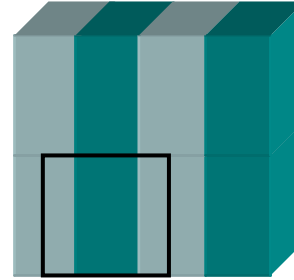
} PSIK network

EPSRC

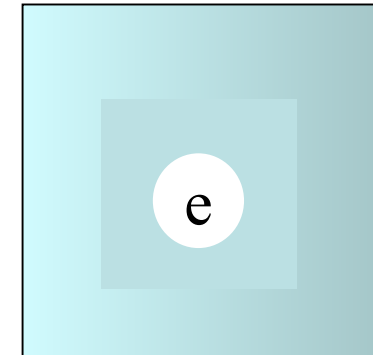
# Ab initio

Periodic boundary conditions

~10-500 atoms →



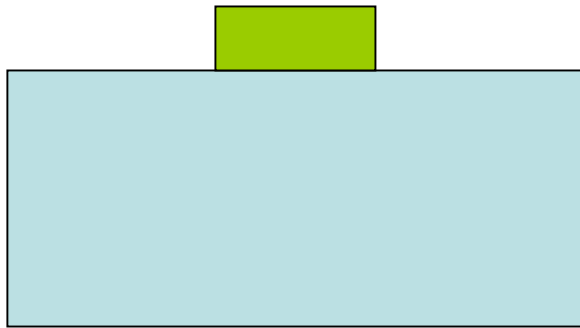
Local density approximation (LDA),  
Generalised Gradient Approximation (GGA)



Codes: CASTEP, VASP, SIESTA, ...

# Growth mode:

## Example of a field of application



Island growth



Layer-by-layer growth

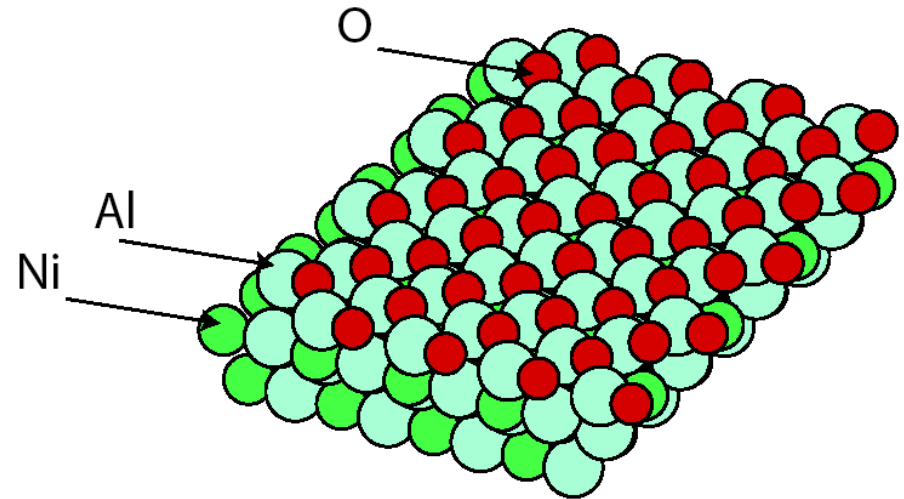
How can we predict growth mode?

- Calculate **nanoscale** thermodynamic driving forces

# Applications and Methodology

## 1.

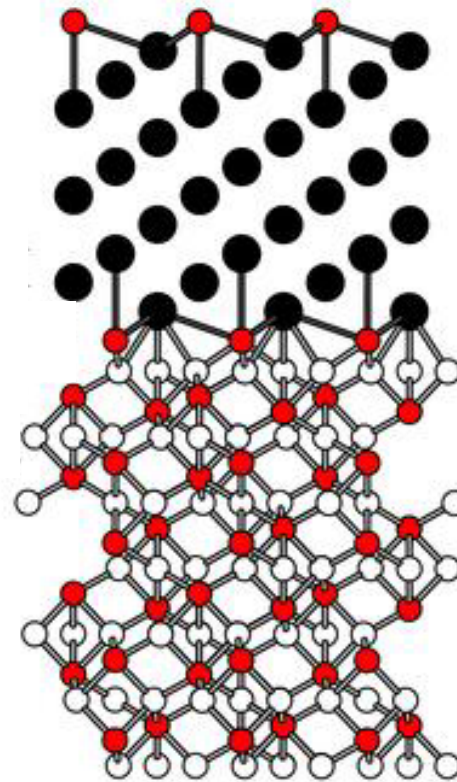
Oxide layer on NiAl(110)



# Applications and Methodology

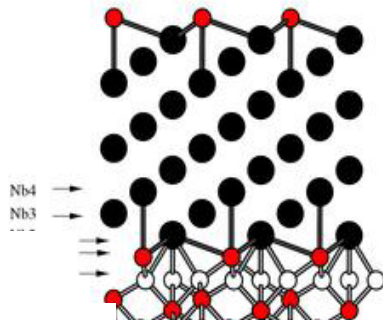
## 2.

Nb(111)-Al<sub>2</sub>O<sub>3</sub>(0001) interface

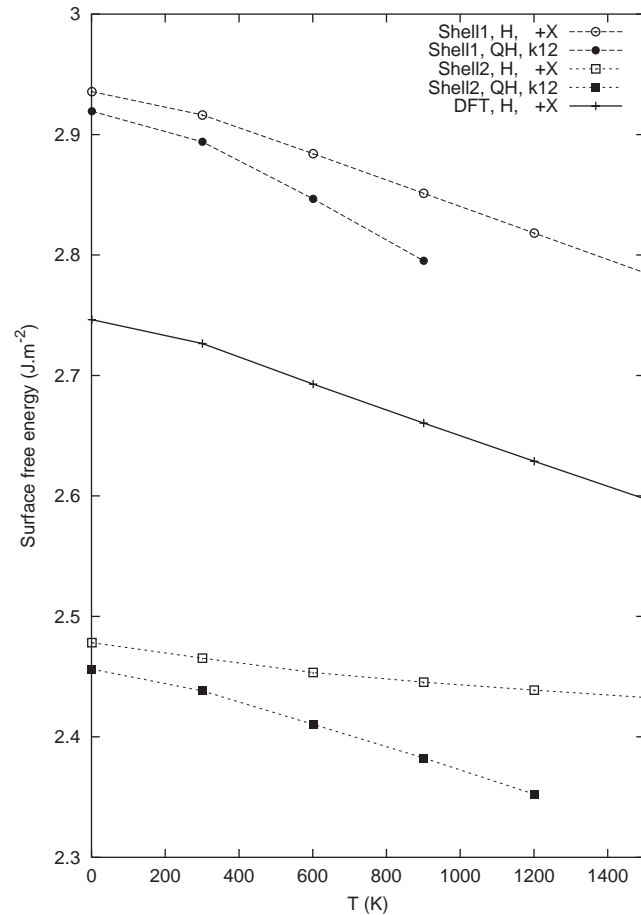


# Applications and Methodology

## 3.



Alumina (0001) surface:  
Energy versus temperature

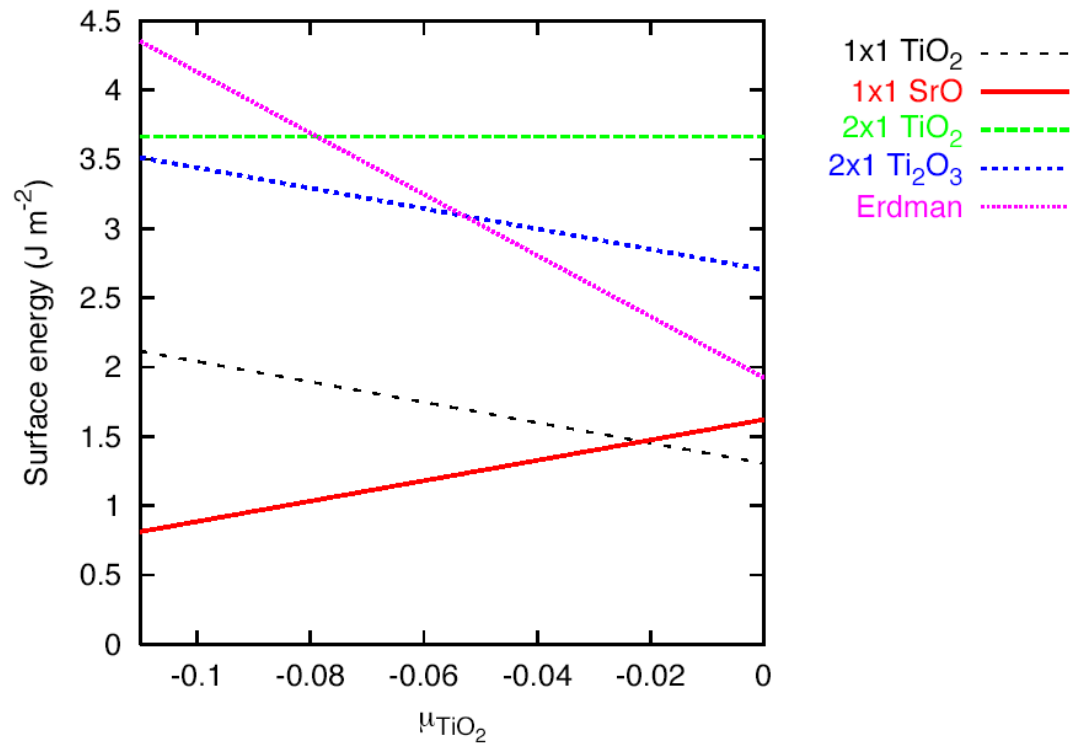




# Applications and Methodology

## 4.

Energy of SrTiO<sub>3</sub>  
surfaces:  
(work in progress)

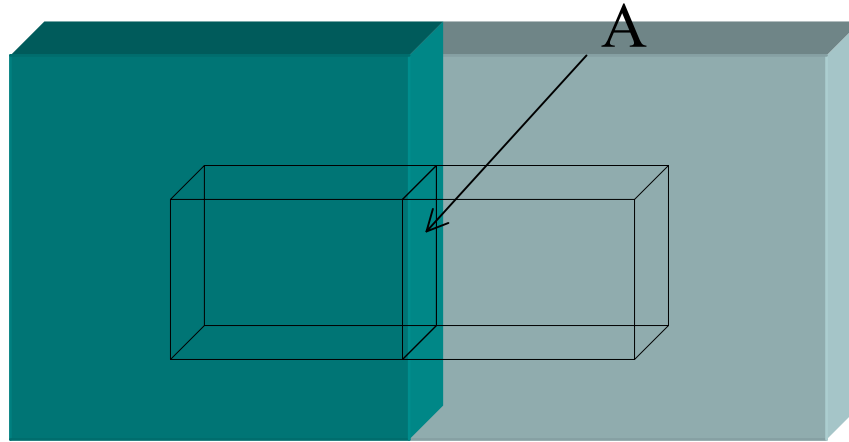


# Reference for interfaces:

Sutton, A. P. and R. W. Balluffi (1995).  
Interfaces in Crystalline Materials. Oxford, Clarendon.

<http://titus.phy.qub.ac.uk>

# Definition of surface energy $\sigma$



For a planar interface, in general:

$$\sigma = (G_s - \sum_i N_i \mu_i) / A$$

Gibbs energy of whole box

Chemical potential of component  $i$

Number of atoms of  $i$

Calculate for different relaxed structures and stoichiometries.  
Minimum  $\sigma$  is predicted equilibrium at given chemical potentials.

# Contributions to surface free energy $\sigma$

## Contributions we would like to know

- Internal energy
- ~~• Configurational entropy of surface atoms~~ — still too difficult!
- Configurational entropy of components in vapour  
(Example: partial pressure of oxygen)
- Configurational entropy of components in solid  
(Example: constitutional defects)
- Vibrational entropy (-TS)

# Thermodynamic setup (NiAl)

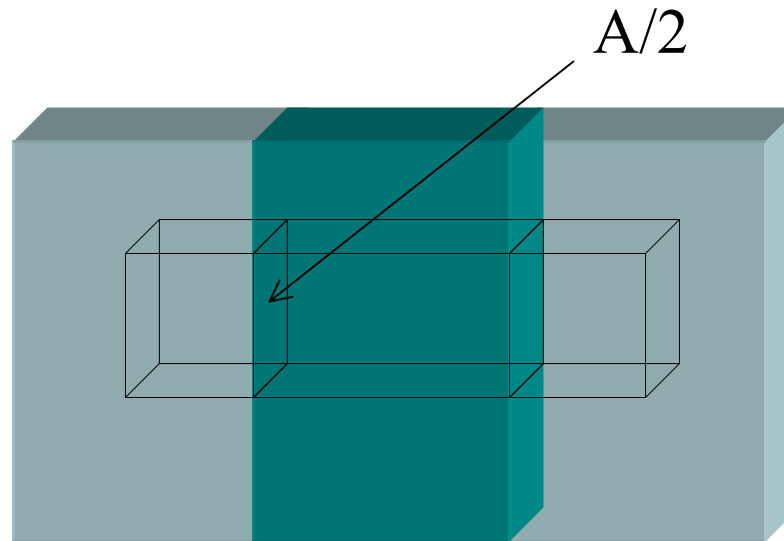
The oxidised surface (represented by a slab in calculations) is in equilibrium with bulk  $\text{Ni}_x\text{Al}_{1-x}$  and vapour with an oxygen partial pressure  $p_{\text{O}_2}$

Phase Rule: Degrees of freedom =  $C - P + 2 = 3$ .

We choose  $x$ ,  $T$  and  $p_{\text{O}_2}$

# Definition of surface energy

## Periodic Boundary Conditions



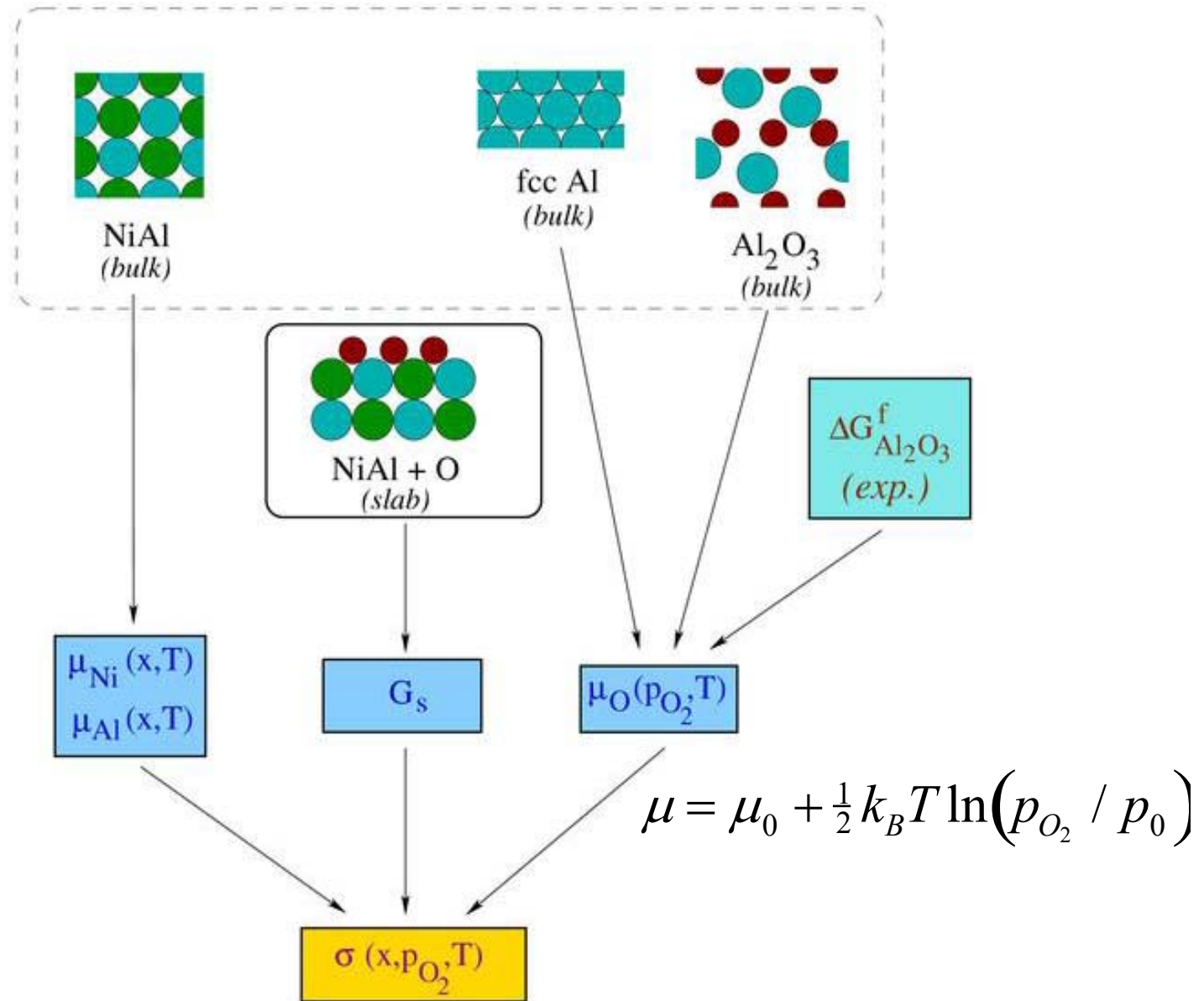
For NiAl:

$$\sigma(x, T, p_{O_2}) = (G_s - N_{Ni} g_{NiAl}) / A - \mu_{Al}(x, T) \Gamma_{Al} - \mu_O(p_{O_2}, T) \Gamma_O$$



$$g_{NiAl} = \mu_{Ni} + \mu_{Al}$$

# Scheme for $\sigma$ calculation



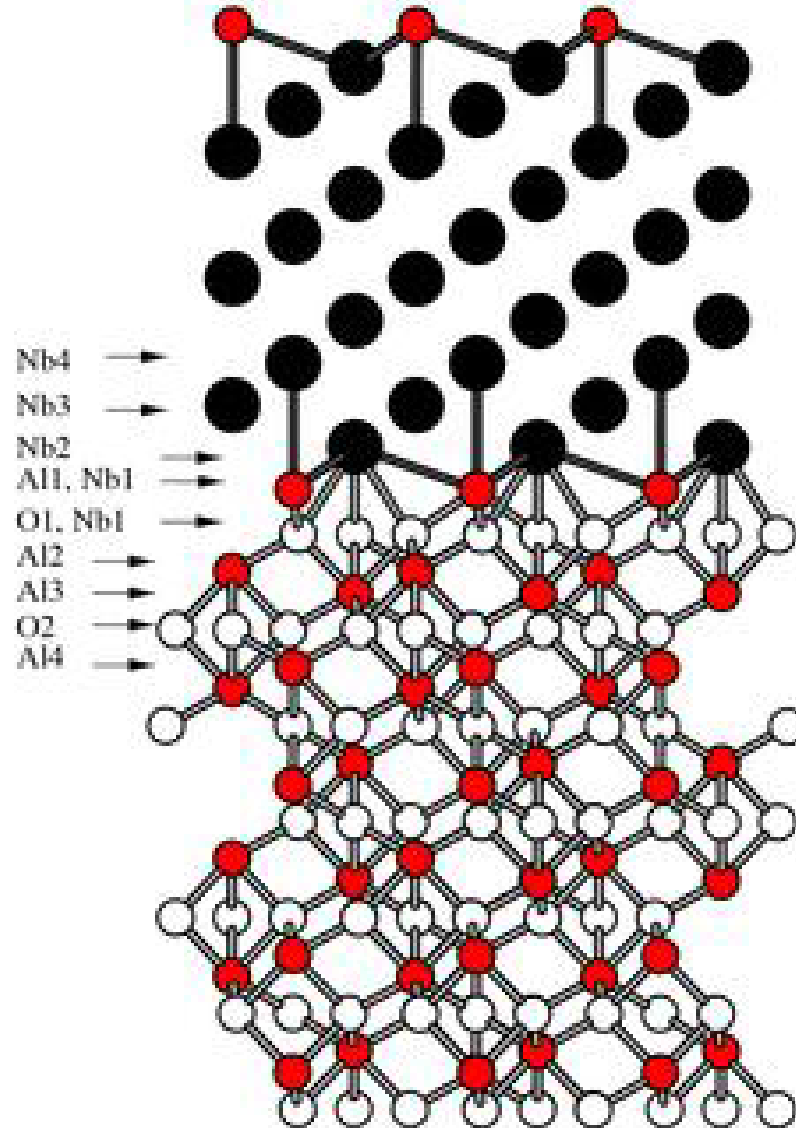
# Change in the point defect energies

	$\Delta E, \text{eV}$			
	bulk	free surface	1/3 ML of O	1 ML of O
Exchange defect: $\text{Ni}_{\text{Al}}^{(2)} + \text{Al}_{\text{Ni}}^{(1)}$	+ 2.66	+1.26	+ 0.42	- 1.87 (!)
$V_{\text{Ni}}$ at the surface:				
bulk $\rightarrow$ 1st layer	—	- 0.13	- 0.52	- 2.96 (!)
bulk $\rightarrow$ 2nd layer	—	+ 0.04		- 0.51

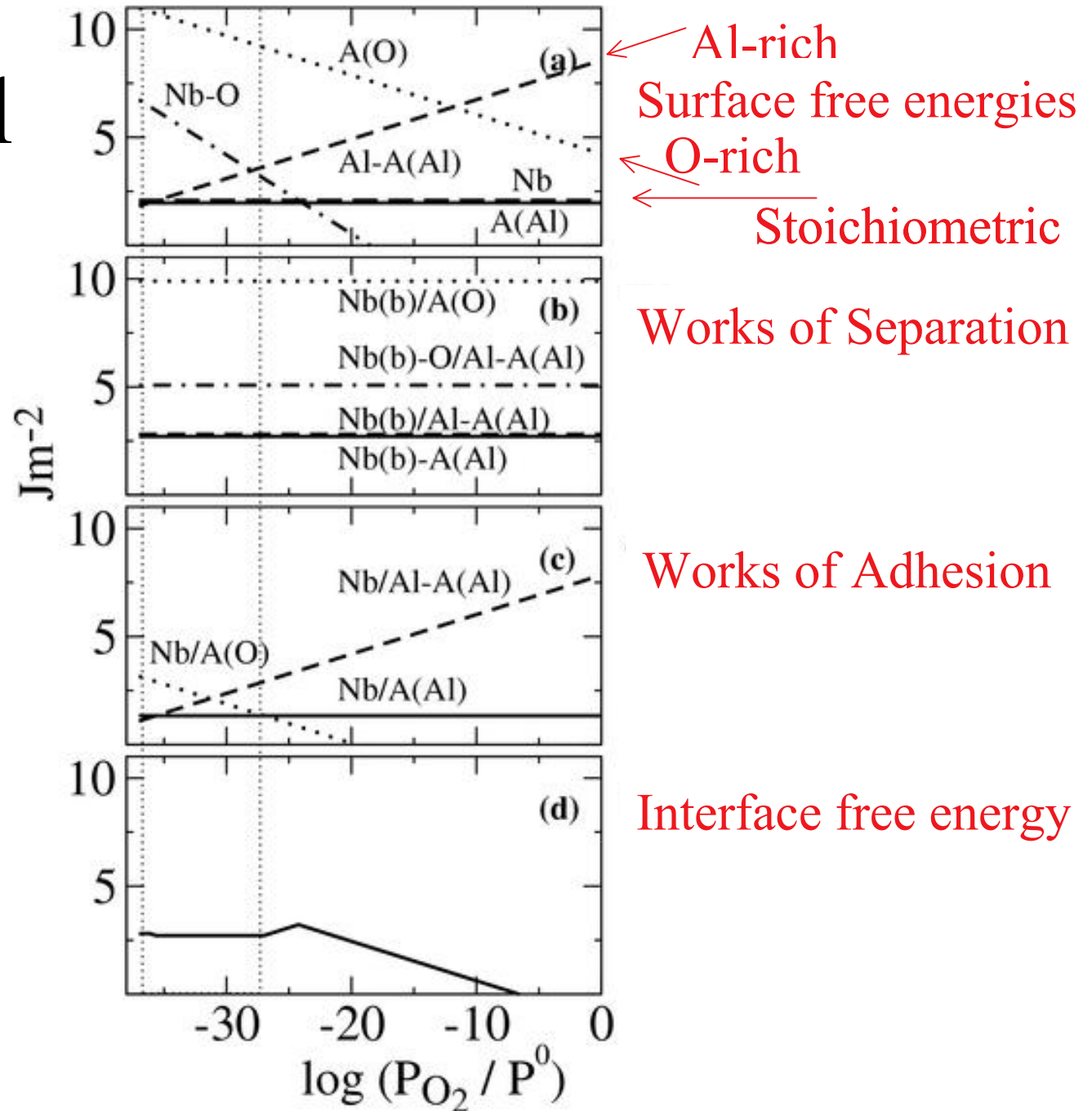
Thus, there is a strong desire of the oxidizing NiAl surface to get rid of Ni in the top layer.



# Nb(111)-Al<sub>2</sub>O<sub>3</sub>(0001) interface



# Interfacial energies



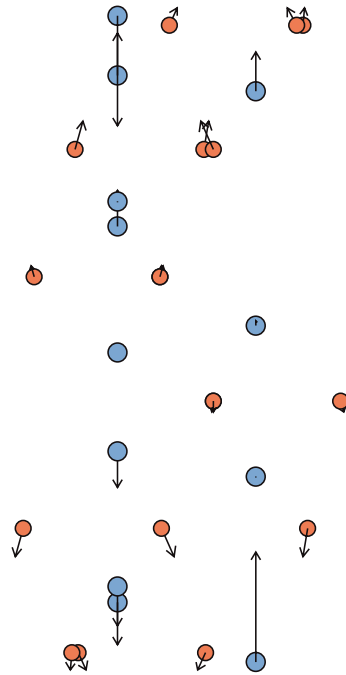
# Vibrational contribution to $\sigma$

Quasiharmonic approximation:

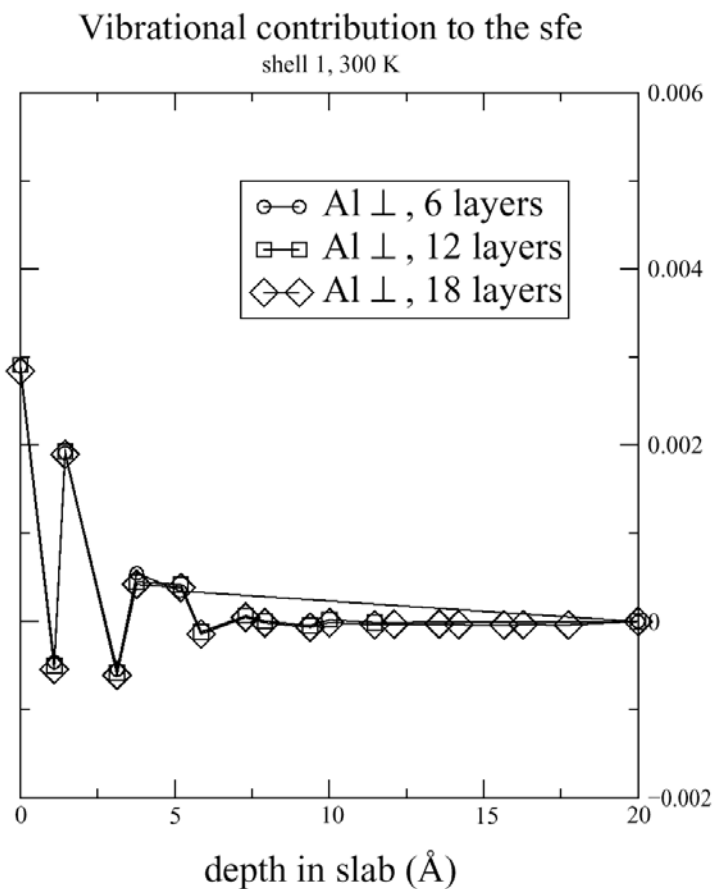
$$F(V, T) = E(V) + \sum_{\mathbf{q}, j} \frac{1}{2} h \nu_j(\mathbf{q}) + kT \ln[1 - \exp(-h \nu_j(\mathbf{q}) / kT)]$$

↑  
Energy of static lattice at volume  $V$

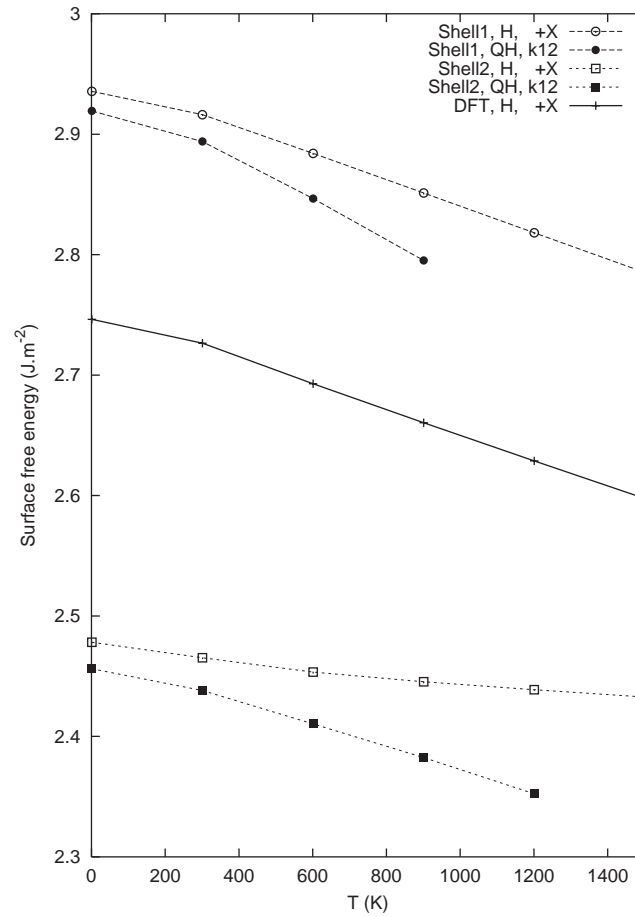
# Surface mode



# Layer-by-layer contributions to vibrational surface free energy



# Surface free energies: Shell models and DFT



# Conclusions

There is a first-principles scheme for calculation of excess surface and interface free energies  $\sigma$  which:

- includes handling excess of components like oxygen in terms of their partial pressure
- includes vibrational entropy

For alumina:

- Al terminated stoichiometric surface stable over most of oxygen pressure range  $< 1$  atmosphere
- Large inward relaxation of Al consistently predicted (70-80%). Experiment gives ca. 50%.....?
- Adequate sampling with 2 phonon k-points
- Ab-initio harmonic: T-dependence of free energy similar to shell model
- Quasiharmonic or better approx. needed for quantitative accuracy ( $\pm 100\text{mJm}^{-2}$  at 1000K)

Thank you for your attention!



# CMMP2003

## 6-9 April

<http://www.qub.ac.uk/mp/con/cmmp2003/>

<http://physics.iop.org/IOP/Confs/CMMP2003/CMMP03.htm>